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## Density Functional Investigation on Alpha-MoO<sub>3</sub> (100): Amines Adsorption and Surface Chemistry

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#### Abstract

The (100) surface of alpha-MoO<sub>3</sub> ( $\alpha$ -MoO<sub>3</sub>) should possess overwhelmingly more exposed Mo than the (010), and the exposed Mo has been extensively considered as active site for amine adsorption. However,  $\alpha$ -MoO<sub>3</sub> (100) has drawn little attention concerning amine sensing mechanism. In this research, adsorption of ammonia (NH<sub>3</sub>), monomethylamine (MMA), dimethylamine (DMA), and trimethylamine (TMA) is systematically investigated by density functional theory (DFT). All of these four molecules have high affinity to  $\alpha$ -MoO<sub>3</sub> (100) through interaction between the N and the exposed Mo, and the affinity is mainly influenced by number of methyl group and number of N-connected H. Adsorption and dissociation of water and oxygen molecule on stoichiometric or defective  $\alpha$ -MoO<sub>3</sub> (100) surface are then simulated to fully understand surface chemistry of  $\alpha$ -MoO<sub>3</sub> (100) in practical condition. At low temperature  $\alpha$ -MoO<sub>3</sub> (100) must be covered with lots of water molecule; the water can desorb or dissociate into hydroxyl groups at high temperature. Oxygen vacancy (Vo) can be generated considering the annealing process during sensor device fabrication; Vo must be filled with O2 molecule, which can further interact with adsorbed water nearby to form hydroxyl groups. According to this research,  $\alpha$ -MoO<sub>3</sub> (100) must be the active surface for amine sensing and its surface chemistry is well understood. In near future, further reaction and interaction will be simulated at  $\alpha$ -MoO<sub>3</sub> (100), and much more attention should be paid to  $\alpha$ -MoO<sub>3</sub> (100) not only theoretically but also experimentally.

#### Key words

DFT; α-MoO<sub>3</sub> (100); amines adsorption; surface chemistry; gas sensing

## **TOC Graph**



#### 1. Introduction

Alpha-MoO<sub>3</sub> ( $\alpha$ -MoO<sub>3</sub>) has drawn intensive attention in gas sensor mainly because of its outstanding sensing performance for amine gases, such as ammonia (NH<sub>3</sub>), trimethylamine (TMA) and triethylamine.<sup>1-7</sup> In our previous research,  $\alpha$ -MoO<sub>3</sub> nanobelts have been prepared and have shown extremely high response and superb selectivity to TMA.<sup>1</sup> For sensing reactions at  $\alpha$ -MoO<sub>3</sub> surface, the TMA was considered to be oxidized by lattice oxygen of  $\alpha$ -MoO<sub>3</sub> into N<sub>2</sub>, CO<sub>2</sub> and H<sub>2</sub>O according to gas chromatography.<sup>1</sup> In term of active site for amines adsorption at  $\alpha$ -MoO<sub>3</sub> surface, it has been extensively believed that the N atom of amines interacts with exposed Mo atom at surface through coordination between the lone-pair electrons of N and empty 4d orbital of Mo.<sup>2-5</sup>



Figure 1 (a) Primitive cell of  $\alpha$ -MoO<sub>3</sub>; primitive cell in (b) ball-stick and (c) polyhedron display style showing four periodic lattices in [100] and [001] directions; (d) bilayer MoO<sub>6</sub> octahedrons demonstrating three structurally distinct oxygen atoms (O<sub>t</sub>, O<sub>as</sub>, and O<sub>s</sub>).

Crystalline MoO<sub>3</sub> has three polymorphs with orthorhombic phase ( $\alpha$ -MoO<sub>3</sub>) being thermodynamically stable. As shown in Figure 1, orthorhombic  $\alpha$ -MoO<sub>3</sub> crystal is comprised of stacked bilayers of distorted

 $MoO_6$  octahedrons through weak van der Waals interaction along the [010] direction, and the octahedrons inside the bilayer connect by sharing vertices along the [100] direction and by sharing edges along the [001] direction.<sup>8</sup> There are three structurally distinct oxygen atoms in each octahedron, i.e. single-coordinated terminal oxygen (O<sub>t</sub>) forming the shortest bond with Mo, the double-coordinated asymmetric oxygen (O<sub>as</sub>) respectively forming one long and one short bond with two Mo, and the triple-coordinated symmetric oxygen (O<sub>s</sub>) forming two equal short bonds with two Mo in the same sublayer and one longest bond with Mo in the other sublayer, as shown in Figure 1d.

The weak interaction between the layers usually results  $\alpha$ -MoO<sub>3</sub> nanomaterials in sheet- or belt-like morphology with the (010) plane being the most exposed surface. Similarly, our previous  $\alpha$ -MoO<sub>3</sub> nanobelts are dominantly exposed with the (010) plane. At the same time, it is worth noting that (100) plane also takes up a large portion for the nanobelt, and (001) can be ignored.<sup>1</sup>

Because of highest percentage of (010) plane at  $\alpha$ -MoO<sub>3</sub> surface,  $\alpha$ -MoO<sub>3</sub> (010) has been experimentally and theoretically investigated for a long period. Figure S1a shows the atomic structure of the  $\alpha$ -MoO<sub>3</sub> (010) surface. It is reasonable that the geometry of the surface (010) is identical to that inside the bulk because all of Ot at (010) plane are singly coordinated with Mo and no bond needs to be broken to form  $\alpha$ -MoO<sub>3</sub> (010) surface.<sup>9</sup> Although Kwak et al. have theoretically demonstrated that NH<sub>3</sub> adsorption on the oxygen vacancy site of  $\alpha$ -MoO<sub>3</sub> (010) surface is much more feasible than stoichiometric one,<sup>2</sup> both experimental and theoretical result have demonstrated that the oxygen vacancy, especially  $O_t$  vacancy ( $V_{Ot}$ ), is unstable or difficult to be generated.<sup>2, 10-11</sup> In early 1983, Firment et al. have studied stoichiometric and oxygen deficient  $\alpha$ -MoO<sub>3</sub> (010) surface, demonstrating that the oxygen deficient surface is unstable.<sup>10</sup> In 2007, Cavalleri et al. have revealed the characteristic of  $V_0$  at  $\alpha$ -MoO<sub>3</sub> (010) by combining angle-resolved near-edge X-ray absorption fine structure (NEXAFS) with density functional theory (DFT), and demonstrated that the relaxation of V<sub>Ot</sub> can make O<sub>as</sub> replace the original O<sub>t</sub>.<sup>11</sup> Some theoretical calculation has verified the result by Cavalleri et al., demonstrating that the O<sub>as</sub> beside the O<sub>t</sub> takes the place of the original Ot if there is one Vot.<sup>12-14</sup> Even though other theoretical research has reported that Oas can stay at its original place when Ot is removed,<sup>15-16</sup> there is little dispute that the formation energy of Vot is so high that it is difficult to generate abundant O<sub>t</sub> vacancies at  $\alpha$ -MoO<sub>3</sub> (010). Therefore, it is not advisable to consider  $\alpha$ -MoO<sub>3</sub> (010) as active surface for amines adsorption.

As mentioned,  $\alpha$ -MoO<sub>3</sub> (100) plane also takes up a large portion of the surface, especially for our nanobelts.<sup>1</sup> Figure S1b shows the atomic structure of  $\alpha$ -MoO<sub>3</sub> (100) surface, which is identically cleaved according to other theoretical research.<sup>9, 13</sup> It is clear that half of the  $\alpha$ -MoO<sub>3</sub> (100) surface is exposed with Mo atoms. According to field emission scanning electron microscope (FESEM) images of our  $\alpha$ -MoO<sub>3</sub> nanobelts, the width/thickness ratio of the nanobelts is even smaller than 10, as shown in Figure S2. Additionally, the formation energy of  $V_{Oas}$  at  $\alpha$ -MoO<sub>3</sub> (100) is lower than that of  $V_{Ot}$  at  $\alpha$ -MoO<sub>3</sub> (010), which means  $V_{Oas}$  is even easier to be generated at  $\alpha$ -MoO<sub>3</sub> (100).<sup>13</sup> Therefore, the  $\alpha$ -MoO<sub>3</sub> (100) must possess overwhelmingly more exposed Mo atom at than the (010) for our nanobelts.

Actually,  $\alpha$ -MoO<sub>3</sub> (100) has already drawn some interest in olefin catalysis field.<sup>17-18</sup> Papakondylis et al. have theoretically explored the adsorption of H<sub>2</sub>O and CO on  $\alpha$ -MoO<sub>3</sub> (100).<sup>9</sup> However, little attention has been paid to amine adsorption at  $\alpha$ -MoO<sub>3</sub> (100). the theoretical research by Papakondylis et al. in early 1996 is still far from understanding the surface chemistry of  $\alpha$ -MoO<sub>3</sub> (100) under gas sensing measurement condition.

In this work, for the first time, amines adsorption at  $\alpha$ -MoO<sub>3</sub> (100) surface is theoretically investigated by density functional calculation. The adsorption energy and charge transfer of NH<sub>3</sub>, monomethylamine (MMA), dimethylamine (DMA), and TMA are calculated and compared. Additionally, to fully understand surface chemistry of  $\alpha$ -MoO<sub>3</sub> (100) under amine sensing measurement condition, the adsorption and dissociation of water and oxygen molecule on stoichiometric or defective  $\alpha$ -MoO<sub>3</sub> (100) surface are simulated.

#### 2. Computational details

Spin-unrestricted DFT calculations within a periodic supercell were performed by Dmol3 package.<sup>19-20</sup> General gradient approximation (GGA) in the form of a Perdew–Burke–Ernzerhof (PBE) was selected for the exchange–correlation functional, and a DFT-D correction was executed to consider van der Waals interaction among bilayers of  $\alpha$ -MoO<sub>3</sub> as well as the weak interaction between gas molecules and  $\alpha$ -MoO<sub>3</sub> surface. The inner electrons of Mo were replaced by an effective core potential (ECP), and electrons of others are in all-electron calculation. The Hubbard parameter, U, was set as 2 eV to treat the coulomb interaction among d-orbital electrons of Mo. A  $9 \times 3 \times 9$  or  $3 \times 3 \times 1$  Monkhorst–Pack grid was respectively used for Brillouin-zone integrations of  $\alpha$ -MoO<sub>3</sub> primitive cell and  $\alpha$ -MoO<sub>3</sub> (100) supercell. For geometry

optimization, convergence tolerance is  $2 \times 10^{-5}$  hartree in energy,  $4 \times 10^{-3}$  hartree Å<sup>-1</sup> in force, and  $5 \times 10^{-3}$  Å in displacement. For transition state (TS) search, a linear synchronous transit/quadratic synchronous transit (LST/QST) method was used with root-mean-square (RMS) convergence of 0.01 hartree/Å.

The adsorption energy ( $E_{ads}$ ), formation energy of  $V_O$  ( $E[V_O]$ ), activation energy ( $E_a$ ) and reaction energy ( $E_{react}$ ) are defined as

$$E[V_{O}] = E_{\alpha-M_{OO3} (100)_{VO}} + 1/2 E_{O2} - E_{\alpha-M_{OO3} (100)}$$
 Equation 1b

- $E_a = E_{TS} E_{reactant}$  Equation 1c
- $E_{react} = E_{procuct} E_{reactant}$  Equation 1d

Herein, in Equation 1a  $E_{\alpha-MoO3 (100)+n gas}$ ,  $E_{\alpha-MoO3 (100)}$  or  $E_{gas}$  represent the total energy of  $\alpha$ -MoO<sub>3</sub> (100) with gas molecules adsorbed, the total energy of  $\alpha$ -MoO<sub>3</sub> (100) or that of a specific gas molecule, and n is the number of adsorbed molecule. Similarly, in Equation 1b,  $E_{\alpha-MoO3 (100)_{-VO}}$  and  $E_{O2}$  represent the total energy of  $\alpha$ -MoO<sub>3</sub> (100) with one oxygen vacancy and that of one oxygen molecule. In Equation 1c and 1d,  $E_{reactant}$ ,  $E_{TS}$  or  $E_{procuct}$  means the energy at the beginning, transition state or end of a reaction. Accordingly, a negative value of  $E_{ads}$  or  $E_{react}$  corresponds to an exothermic adsorption or reaction. To analyze the charge transfer between the surface and adsorbed molecules, Mulliken charge was calculated. The unit of Mulliken charge is the absolute value of elementary electron charge, thus the positive value of adsorbed molecule means electron donation from the molecule to  $\alpha$ -MoO<sub>3</sub> surface.

The optimized orthorhombic  $\alpha$ -MoO<sub>3</sub> possesses lattice parameters of a = 3.95 Å, b = 13.86 Å and c = 3.74 Å, which is very close to experimental value of a = 3.96 Å, b = 13.86 Å and c = 3.69 Å.<sup>13, 21</sup> Based on the optimized  $\alpha$ -MoO<sub>3</sub> primitive cell, a (1 × 4) supercell of  $\alpha$ -MoO<sub>3</sub> (100) surface was cleaved, as shown in **Figure S1b and Figure S2**. The supercell is in size 13.86 × 14.95 × 37.36 Å<sup>3</sup> with 240 O atoms and 80 Mo atoms in total. A 15 Å vacuum layer is introduced to avoid interaction among the slabs in [100] direction and some bottom atoms were fixed to simulate bulk atoms, which is red color in **Figure S3**. The  $\alpha$ -MoO<sub>3</sub> (100) surface is not smooth but with terrace and channel because of interlaced stacking of octahedron bilayers along [010] direction. To clearly show the terrace-channel peculiarity of the  $\alpha$ -MoO<sub>3</sub> (100) surface, the supercell is presented with two periodic lattices along [010] direction.

#### 3. Results and discussion

#### Amines adsorption at stoichiometric $\alpha$ -MoO<sub>3</sub> (100) surface

With assurance of  $\alpha$ -MoO<sub>3</sub> (100) being active surface for amines adsorption, DFT calculation was performed to investigate adsorption site, adsorption energy and charge transfer for four amines, i.e. NH<sub>3</sub>, MMA, DMA, and TMA. Figure 2a and b shows adsorption configurations of NH<sub>3</sub> at terrace of  $\alpha$ -MoO<sub>3</sub> (100). It is similar to what has been believed that N atom bonds to the exposed Mo atom by offering its lone-pair electron to the empty 4d orbital of Mo. Additionally, two H atoms of NH<sub>3</sub> may have hydrogen bond interaction with adjacent  $O_{as}$ . The adsorption energy is -1.47 eV, suggesting high ability of  $\alpha$ -MoO<sub>3</sub> (100) to adsorb NH<sub>3</sub>. Figure 2c-h shows the adsorption configurations of MMA, DMA, and TMA at terrace of α-MoO<sub>3</sub> (100). All of these adsorptions mainly originate from the interaction between the N and exposed Mo. Nevertheless, there are still some differences. As listed in Table 1, comparisons are made on the adsorption energy, the N-Mo distance and the charge transfer of these amines. For the adsorption at terrace,  $E_{ads}$  (in absolute value) increases with the number of methyl groups increasing from 0 to 2, i.e. from NH<sub>3</sub> to DMA, while E<sub>ads</sub> decreases when it goes to 3, i.e. TMA. The increase from NH<sub>3</sub> to DMA can be mainly ascribed to electron-releasing effect of methyl group. The N atom is more electron-abundant in the amine molecule with more methyl group, which renders N atom more affinitive to Lewis site of Mo. Additionally, heavier molecule somehow facilitates van der Waals interaction. In term of the Eads reduction from DMA to TMA, it can be ascribed to the three methyl groups which have a repulsive effect with the surface bonds and prevent the N to coordinate with Mo. Hence, the distance between the N and Mo is 2.558 Å, longer than that of the others, as shown in Table 1. Furthermore, the low E<sub>ads</sub> also results from the lack of hydrogen bond interaction between TMA, which is has no H atom connected to N, and the surface Oas.

The repulsion effect between methyl group and Mo-O<sub>t</sub> bond can be verified by comparing two MMA adsorption configurations. **Figure S4** shows MMA adsorption at terrace of  $\alpha$ -MoO<sub>3</sub> (100) without or with N–C bond overlapping with Mo–O<sub>t</sub> bond. The angle between N–C bond and N–Mo bond ( $\angle$ CNMo) and the angle between N–Mo bond and Mo–O<sub>t</sub> bond ( $\angle$ NMoO<sub>t</sub>) is respectively 118.35° and 82.60° in the configuration of **Figure S4a and S4b**. Comparatively,  $\angle$ CNMo and  $\angle$ NMoO<sub>t</sub> is respectively 118.79° and 86.54° in the configuration of **Figure S4c and S4d**. The larger angles, especially for the  $\angle$ NMoO<sub>t</sub>, demonstrate a more severe repulsion in the latter configuration. The adsorption energy for the former (-1.70 eV) is slightly higher than its counterpart (-1.68 eV) despite that one less hydrogen bond can be

formed. Accordingly, the repulsion effect between methyl group and Mo–O<sub>t</sub> plays a more important role than the hydrogen bond interaction for the MMA adsorption. It is also for this reason that the N–C bond does not overlap with the Mo–O<sub>t</sub> bond for the adsorption of DMA and TMA, as shown in Figure 2e-h.



**Figure 2** Adsorption configuration at terrace of α-MoO<sub>3</sub> (100): (a, b) NH<sub>3</sub>, (c, d) MMA, (e, f) DMA, and (g, h) TMA.

**Table 1** Adsorption energy, distance between the N and the Mo, and Mulliken charge for amines adsorbed at terrace or channel site of  $\alpha$ -MoO<sub>3</sub> (100).

Amines		NH <sub>3</sub>	MMA	DMA	TMA
Number of methyl group		0	1	2	3
$\mathbf{E}$ (- $\mathbf{V}$ )	Terrace	-1.47	-1.70	-1.77	-1.37
$\mathbb{L}_{ads}(\mathbf{ev})$	Channel	-1.66	-1.80	-1.69	-1.36
N-Mo (Å)	Terrace	2.343	2.345	2.401	2.558
	Channel	2.338	2.346	2.442	3.934
Mulliken charge ( e )	Terrace	0.283	0.308	0.319	0.271
	Channel	0.285	0.301	0.316	0.498

Besides terrace, amine molecules can also be adsorbed at channel, as shown in Figure S5. The adsorption energy is also listed in Table 1. By comparing the  $E_{ads}$  at terrace and channel site, it can be concluded that small amine molecules, such as NH<sub>3</sub> and MMA, tend to be adsorbed at the channel, while

the large molecules, such as DMA and TMA, at terrace. Interestingly,  $E_{ads}$  of TMA at channel is very approaching to that at terrace, even though the N is still distant from the Mo (3.934 Å). The reason can be that more H atoms in the methyl groups can interact with more  $O_{as}$  and  $O_{t.}$  at the surface when TMA is at channel.

The Mulliken charge is also analyzed for each configuration, as listed in **Table 1**. Firstly, they are all positive, suggesting electron donation from adsorbed amines to  $\alpha$ -MoO<sub>3</sub> (100) surface. Secondly, comparing the Mulliken charge of the four amines at terrace, we find that DMA donates electron at most, followed by MMA, NH<sub>3</sub> and TMA, which shows the same trend as E<sub>ads</sub>. Nevertheless, for TMA at channel, the Mulliken charge reaches 0.498 |e|, much higher than others. As mentioned before, it can be attributed to the interaction between the H atoms in the methyl group with the surface O<sub>as</sub> and O<sub>t</sub>. Except TMA, the Mulliken charge at channel is slightly higher than that at terrace.

Based on above result, the amine molecules are highly affinitive to  $\alpha$ -MoO<sub>3</sub> (100) surface and can donate many electrons once adsorbed. This somehow can explain why the resistance of n-type  $\alpha$ -MoO<sub>3</sub> decreases when amine gas is injected.<sup>1-7, 22</sup> However, adsorption is only the first step for gas sensing reaction, and continuous dissociation or reaction at the surface can be more significant for understanding sensing mechanism.

#### $O_2$ , $H_2O$ , and $V_0$ at $\alpha$ -Mo $O_3$ (100) surface

Instead of continuing to simulate decomposition or reaction of these adsorbed amines, we would like to investigate the surface chemistry of at  $\alpha$ -MoO<sub>3</sub> (100) with taking O<sub>2</sub>, H<sub>2</sub>O and V<sub>0</sub> into consideration. It must be kept in mind that there is never a perfect surface. It is necessary to consider whether and how can Mo sites be exposed under gas sensing measurement condition, in which a large number of O<sub>2</sub> and H<sub>2</sub>O molecules must interact with the surface. O<sub>2</sub> concentration in air is about 21 % (210,000 ppm) and H<sub>2</sub>O concentration at 50 % relative humidity at 25 °C is about 10,000 ppm,<sup>23</sup> which are both much higher than amine concentration in sensing measurement.

The effect of the oxygen molecule in air cannot be excluded for understanding gas sensing mechanism. For some metal oxides, reducing gas is believed to be oxidized by adsorbed oxygen, but for some others, it is lattice oxygen.<sup>24-29</sup> Lattice oxygen of  $\alpha$ -MoO<sub>3</sub> is abundant and active, and some have believed it is lattice oxygen that oxidize reducing gas molecules,<sup>1,6</sup> while others adsorbed oxygen.<sup>5,7</sup> Hence, the O<sub>2</sub> adsorption at stoichiometric  $\alpha$ -MoO<sub>3</sub> (100) is simulated, as shown in **Figure S6**. It is found that E<sub>ads</sub> is 0.57 eV and 0.47 eV for adsorption at terrace and channel site of  $\alpha$ -MoO<sub>3</sub> (100) and O<sub>2</sub> molecule is still of ~3 Å distance from the exposed Mo site. Therefore, O<sub>2</sub> adsorption on stoichiometric  $\alpha$ -MoO<sub>3</sub> (100) surface is not favorable.

As mentioned before, humidity has also to be considered. It has been discovered that H<sub>2</sub>O molecule has huge effects on gas sensing properties of metal oxides, and certainly for α-MoO<sub>3</sub>.<sup>1, 30-32</sup> Water adsorption at  $\alpha$ -MoO<sub>3</sub> (100) has already been theoretically investigated, and it has found that E<sub>ads</sub> is about -1 eV for full coverage of H<sub>2</sub>O (each exposed Mo with one H<sub>2</sub>O molecule).<sup>9</sup> Herein, H<sub>2</sub>O adsorption at  $\alpha$ -MoO<sub>3</sub> (100) in different coverages (1/8, 1/4, 1/2 and full) as well as its dissociation are simulated, as shown in Figure 3. The adsorption energy and Mulliken charge of these configurations are listed in Table 2. The result demonstrates that water molecule tends to be adsorbed at channel site at low coverage (1/8 and 1/4) with its O atom bonding to the exposed Mo and its two H atoms having hydrogen bond interaction with surface O<sub>as</sub>, as shown in Figure 3a-d. When the coverage increases to 1/2, the water molecule cannot be entirely adsorbed at channel, and half of them terrace, as shown in Figure 3e. Figure 3f shows the relaxed configuration of four water molecules at channel site, and it can be found that some of the water molecules rotate a little to a lower-energy state in which one H can interact with Oas at channel and the other H can interact with Ot at terrace. This suggests that the hydrogen bond interaction between H atom and the Oas at channel is reduced if there are eight H atoms from four water but only four Oas. Moving half of water molecules to terrace can render stronger hydrogen bond interaction. Figure S7 shows other two possible configurations of 1/2 water, which are also less stable that Figure 3e. The rotation of water at channel is also found for fully covered  $\alpha$ -MoO<sub>3</sub> (100), as shown in Figure 3g and h. The configuration in Figure 3g with three water rotating is somewhat more stable than that in Figure 3h with only one rotating molecule. Notably, when the three rotate, the remaining molecule does not need to rotate because it can have full hydrogen bond interaction with two Oas at channel. None of four water molecules at terrace rotate because none of Ot from channel is at same height as the H atoms and to interact with them. As for the Mulliken charge, they are all positive, meaning the water adsorption can also result in resistance reduction for n-type  $\alpha$ -MoO<sub>3</sub>, which has already been verified by our previous research.<sup>1</sup> From Table 2, it can also be seen that less electronic effect is induced as the number of adsorbed water molecules increases.



Figure 3 Water adsorption at  $\alpha$ -MoO<sub>3</sub> (100) in (a, b) 1/8, (c, d) 1/4, (e, f) 1/2 and (g, h) full coverage: (a) At channel and (b) at terrace site; (c) two both at channel and (d) one at channel and the other terrace; (e) all at channel and (f) two at channel and the others at terrace; (g, h) the water molecules adsorbed at channel in different configuration.

•	Coverage	1/8	1/4	1/2	Full
E <sub>ads</sub> (eV)	$\mathbf{E}_{(\mathbf{a}V)}$	-1.20 <sup>a</sup>	-1.19 <sup>c</sup>	-1.16 <sup>e</sup>	-1.14 <sup>g</sup>
	$E_{ads}(ev)$	-1.13 <sup>b</sup>	$-1.17^{d}$	$-1.15^{f}$	$-1.11^{h}$
Mu	M11:1	0.192 <sup>a</sup>	0.187 <sup>c</sup>	0.184 <sup>e</sup>	0.174 <sup>g</sup>
	Mulliken charge ( e )	0.182 <sup>b</sup>	0.187 <sup>d</sup>	0.169 <sup>f</sup>	0.175 <sup>h</sup>

**Table 2** Average adsorption energy and Mulliken charge for one water molecule at  $\alpha$ -MoO<sub>3</sub> (100) in different coverage (Superscripts of a-h represents the configuration in Figure 3a-h.)

The adsorbed water may also dissociate when temperature is elevated. Herein, dissociation of one water molecule at terrace or channel of  $\alpha$ -MoO<sub>3</sub> (100) is simulated. As clearly shown in **Figure 4**, the molecularly adsorbed water is more stable than dissociated water either at terrace or at channel according to the relative energy and energy barrier. Nonetheless, the water molecule at channel takes more possibility to

dissociate with  $E_a$  and  $E_{react}$  of 0.78 eV and 0.57 eV, compared with 1.05 eV and 0.76 eV at terrace. From the atomic structures on the right, it can be seen that the reason why the dissociated water is more stable at the channel than at the terrace is that the formed HO<sub>as</sub> group of the former can have hydrogen bond interaction with nearby O<sub>t</sub> in terrace with a distance of 2.044 Å.



Figure 4 Water dissociation process at terrace and channel of  $\alpha$ -MoO<sub>3</sub> (100). The left shows relative energy at each process with reference to overall energy of stoichiometric surface plus vapor water molecule, and the right shows atomic structure in different process.

Based on the calculation of water adsorption, it can be deduced that there must be numerous water molecules previously adsorbed at  $\alpha$ -MoO<sub>3</sub> (100) in measurement condition, especially at room or relatively low temperature. The adsorption site for water is same as that for amines, thus the existence of water should prevent the adsorption of amines. However, if working temperature is elevated, some of water molecules may be desorbed, which can provide site for amine adsorption. The water at the terrace is more likely to desorb because the E<sub>ads</sub> at terrace is somewhat lower than that at channel. In contrast, the molecules at channel the may either be desorbed or dissociate.

Besides  $O_2$  and  $H_2O$ ,  $V_O$  must also be taken into consideration, because the annealing process during material synthesis and device fabrication probably leads to  $V_O$  formation.<sup>1</sup> Figure 5a shows five kinds of oxygen at  $\alpha$ -MoO<sub>3</sub> (100) surface.  $O_{as}(t)$  and  $O_{as}(c)$  represent  $O_{as}$  at terrace and channel, and  $O_s(t)$  represents

O<sub>s</sub> at terrace. Notably, there are two kinds of O<sub>t</sub> even only at the terrace, one of them is connected with fully coordinated Mo, which is labeled O<sub>t</sub>(tf), and the other is connected with exposed Mo, which is labeled O<sub>t</sub>(te). Figure 5b-f shows the relaxed configuration of α-MoO<sub>3</sub> (100) with five different vacancies, i.e.  $V_{Oas(t)}$ ,  $V_{Oas(c)}$ ,  $V_{Ot(tf)}$ ,  $V_{Ot(te)}$ , and  $V_{Os(t)}$ , whose vacancy formation energy is respectively 1.23 eV, 1.30 eV, 1.23 eV, 1.35 eV, and 3.75 eV. Interestingly,  $E[V_{Oas(t)}]$  is same as the  $E[V_{Ot(tf)}]$ , and the underlying reason is that α-MoO<sub>3</sub> (100) with one  $V_{Ot(tf)}$  is relaxed to the same configuration as that with one  $V_{Oas(t)}$ , as can be seen in Figure 5b and d. Movie S1 shows the process more clearly. This process is similar to what has happened at α-MoO<sub>3</sub> (010) in other theoretical research.<sup>11-14</sup> Therefore,  $V_{Oas(t)}$  is most likely to generate. The relaxed configuration with  $V_{Ot(te)}$  is also very interesting, and Figure 5e and Movie S2 show that the exposed Mo is dragged inside by two O<sub>as</sub> underneath.



Figure 5 (a)  $\alpha$ -MoO<sub>3</sub> (100) surface showing five kinds of oxygen, and (b-f) the relaxed configuration of  $\alpha$ -MoO<sub>3</sub> (100) with five different vacancies.

The  $V_{Oas(t)}$  may form during annealing process, whereas it has a high tendency to interact with O<sub>2</sub> and H<sub>2</sub>O in air after the annealing. **Figure S8** shows adsorption and dissociation process of O<sub>2</sub> at  $\alpha$ -MoO<sub>3</sub> (100) with one  $V_{Oas(t)}$ . Compared with stoichiometric surface, O<sub>2</sub> can be easily adsorbed at  $V_{Oas(t)}$  with E<sub>ads</sub> of -0.63 eV. Once O<sub>2</sub> is adsorbed, the O–O bond elongates from 1.233 Å to 1.363 Å, and Mulliken charge of

 $O_2$  becomes -0.443 |e|, which suggests electron extraction from surface. This implies the tendency of  $O_2$  to dissociate. However,  $E_a$  and  $E_{react}$  for its dissociation are as high as 1.64 eV and 1.18 eV. It suggests  $O_2$  is hard to dissociate, herein, especially at low temperature. A water molecule may also be adsorbed at or around  $V_{Oas(t)}$  site, while the calculation result shows that  $V_{Oas(t)}$  does not facilitate adsorption of water. As shown in **Figure S9**, the most favorable adsorption configuration is **Figure S9e and f**, with  $E_{ads}$  of -1.06 eV, which is still a little lower than that at stoichiometric surface.



Figure 6 Adsorption and dissociation of both  $O_2$  and  $H_2O$  at  $\alpha$ -MoO<sub>3</sub> (100) with one  $V_{Oas(t)}$ . The left shows relative energy at each process with reference to overall energy of defective surface plus vapor  $O_2$  and  $H_2O$  molecule, and the right shows atomic structure in different process.

Co-adsorption of  $O_2$  and  $H_2O$  around  $V_{Oas(t)}$  is also of high possibility, because there are three exposed Mo atoms very close to each other. **Figure 6** shows co-adsorption and dissociation of both  $O_2$  and  $H_2O$  at  $\alpha$ -MoO<sub>3</sub> (100) with one  $V_{Oas(t)}$ . The energy for the co-adsorption is -1.59 eV, suggesting the high feasibility, even though it is a little lower than the sum of  $E_{ads}$  of single  $O_2$  (-0.63 eV) and single  $H_2O$  (-1.06 eV) adsorption at or around  $V_{Oas(t)}$ . Additionally, the presence of  $H_2O$  facilitates the dissociation of  $O_2$  with  $E_a$ and  $E_{react}$  of 1.57 eV and 1.03 eV, which is respectively 0.07 eV and 0.15 eV lower than those with absence of  $H_2O$ . The dissociation of  $O_2$  in turn facilitates dissociation of  $H_2O$ . The H–O bond closest to dissociated  $O_2$  elongates to 0.995 Å, and the other 0.990 Å. The former is easier to rupture by transferring the H to the  $O_{as}$ , which is dissociated from  $O_2$ , with  $E_a$  and  $E_{react}$  of 0.83 eV and 0.34 eV. The  $E_a$  and  $E_{react}$  are 0.22 eV and 0.42 eV lower than those of water dissociation at terrace of stoichiometric  $\alpha$ -MoO<sub>3</sub> (100). The HO<sub>as</sub> is not stable, and the H can be easily ( $E_a = 0.23$  eV) captured by the other O atom dissociated from  $O_2$ , reaching a more stable configuration. Finally,  $\alpha$ -MoO<sub>3</sub> (100) surface with two terminal hydroxyl groups is formed.

Therefore, in practive,  $\alpha$ -MoO<sub>3</sub> (100) must already be covered with lots of water molecule, especially at low temperature. If the temperature is high enough, adsorbed water may dissociate to form a terminal HO and rooted HO<sub>as</sub> group. Because of the annealing process, V<sub>Oas(t)</sub> is likely to be generated, but can be easily filled with O<sub>2</sub> molecule. The adsorbed O<sub>2</sub> molecule at V<sub>Oas(t)</sub> may dissociate with the nearby H<sub>2</sub>O, and this finally leads to two terminal hydroxyl groups at surface.

#### 4. Conclusion

It has been rationally analyzed that  $\alpha$ -MoO<sub>3</sub> (100) should possess overwhelmingly more exposed Mo than  $\alpha$ -MoO<sub>3</sub> (010) based on previous experimental and theoretical investigation. Accordingly, in this research, amines adsorption as well as interaction between O<sub>2</sub>, H<sub>2</sub>O and V<sub>O</sub> at  $\alpha$ -MoO<sub>3</sub> (100) surface are systematically investigated by density functional theory calculation. All four amine molecules have high affinity to the  $\alpha$ -MoO<sub>3</sub> (100) surface through interaction between the N and the exposed Mo, while the affinity is mainly influenced by number of methyl group, number of N-connected H. Regarding the surface chemistry,  $\alpha$ -MoO<sub>3</sub> (100) under practical conditions must be covered with many water molecules at low temperature; besides, O<sub>2</sub> molecule filling V<sub>Oas</sub> and hydroxyl group can also be present at  $\alpha$ -MoO<sub>3</sub> (100) under practical gas sensing measurement if considering interaction between O<sub>2</sub>, H<sub>2</sub>O and V<sub>0</sub>.

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### Supporting Information

# Surface chemistry of α-MoO<sub>3</sub> (100) and its active site for amines adsorption through density functional calculation

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Figure S1 (a)  $(4 \times 4)$  supercell of  $\alpha$ -MoO<sub>3</sub> (010) surface, (b)  $(1 \times 4)$  supercell of  $\alpha$ -MoO<sub>3</sub> (100) surface showing two periodic lattice in [010] direction.



Figure S2 Field emission scanning electron microscope (FESEM) images of our  $\alpha$ -MoO<sub>3</sub> nanobelts



Figure S3  $\alpha$ -MoO<sub>3</sub> (100) slab with a 15 Å vacuum layer (The relaxed atoms is in grey and the fixed in red.)



Figure S4 Two configurations for MMA adsorption at terrace of  $\alpha$ -MoO<sub>3</sub> (100): (a, b) N–C bond not overlapping with Mo–O<sub>t</sub> bond and (c, d) N–C bond overlapping with Mo–O<sub>t</sub> bond in the view from [100] direction.



Figure S5 Adsorption configuration at channel of  $\alpha$ -MoO<sub>3</sub> (100): (a, b) NH<sub>3</sub>, (c, d) MMA, (e, f) DMA, and

(g, h) TMA.



Figure S6  $O_2$  adsorption configuration at  $\alpha$ -MoO<sub>3</sub> (100): (a, b) Terrace and (c, d) channel.



**Figure S7** Two other configurations of 1/2 coverage of water: (a) All of four water molecules are adsorbed at terrace, (b) one at terrace and three at channel.

The  $E_{ads}$  for Figure S7a and S7b is respectively -1.05 eV and -1.15 eV, compared with -1.16 eV of Figure 3e.



Figure S8 Adsorption and dissociation process of  $O_2$  at  $\alpha$ -MoO<sub>3</sub> (100) with one  $V_{Oas(t)}$ . The left shows relative energy at each process with reference to overall energy of defective surface and vapor  $O_2$  molecule, and the right shows atomic structure in different process.

Because of surface symmetry, adsorption configuration by interacting with Mo1 and Mo2 is same as that with Mo1 and Mo3.



Figure S9 Water adsorption at  $\alpha$ -MoO<sub>3</sub> (100) with one V<sub>Oas(t)</sub>: (a, b) at vacancy and (c-f) beside vacancy but a little different in configuration.

The  $E_{ads}$  for (a, b), (c, d) and (e, f) is respectively -0.96 eV, -1.00 eV, and -1.06 eV.